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(54) An aluminum alloy composite material for brazing.

(57) The composite material has a structure comprising: a core member(1) of an aluminum alloy, the content of Mg being restricted to 0.2 wt.% or less as an impurity, consisting essentially of 0.2 to 1.5 wt.% of Mn, 0.3 to 1.3 wt.% of Si, 0.02 to 0.3 wt.% of Ti, and, as required, 0.6 wt.% or less of Cu, 0.3 wt.% or less of Cr and 0.2 wt.% or less of Zr; an Al-Si filler member(3) clad on one surface of the core member; and a cladding member(2) of an aluminum alloy clad on the opposite side of said core member 1, consisting essentially of 0.3 to 3 wt.% of Mg and as required, 5 wt.% or less of Zn. The thickness of the core member(1) is preferably, 2.5 times or more greater than that of the filler member(3) falling within a range of 0.1 to 1 mm. With respect to a tube member for a radiator which is assembled by the brazing method with noncorrosive flux, the pitting potential of the core member (positive) is higher than that of the cladding member and is higher than that of the filler member, and the differences in pitting potential between the core member and the cladding member and between the core member and the filler member are 30 to 120 mV. With this, it is possible to obtain a tube member for a radiator which has excellent corrosion resistance for a long period of time.

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The present invention relates to an aluminum alloy composite material for brazing, which is improved in strength, corrosion resistance and brazability, and which provides remarkable advantages when used for assembly process by brazing method with noncorrosive flux.

Aluminum alloy composite materials for brazing are used for a tube member of a brazed radiator, and
 5 the like, and conventionally, brazing sheets comprising an Al-Mn alloy (JIS A3003) core member and an Al-Si filler member clad on the core member are used for this purpose. The brazing sheet with the A3003 core member, however, has a post-brazing strength of not greater than 12 kgf/mm² or thereabouts and is not sufficiently resistant to corrosion.

If Mg is added to the aluminum alloy for the core member, the strength of the composite material can
 10 be improved, but this results in an increased erosion of the core member due to the filler member and also in a reduction of brazability and corrosion resistance. Particularly, in the case of employing brazing method with noncorrosive flux, the Mg content of the core member in excess of 0.2% entails a substantial drop of the brazability, and accordingly, the addition of Mg to the core member of the aluminum alloy must be restricted.

15 There has been also presented, in Japanese Published Unexamined Patent Application No.175093/1990, an aluminum alloy composite material for brazing improved in the post-brazing strength without inhibiting brazability. This composite material, however, cannot achieve a satisfactory corrosion resistance when the material is made to be drawn thinner.

Thus, with conventional techniques, it is difficult to obtain a tube member of a brazed radiator which has
 20 high strength and high corrosion resistance and yet is excellent in brazability. As for heat exchanger such as an automobile radiator, thin materials are needed to reduce the weight and cost, and accordingly, there is a demand for an aluminum alloy composite material for brazing which has high strength and high corrosion resistance and is excellent in brazability, but no such material has been developed yet.

It is an object of the present invention to provide an aluminum alloy composite material for brazing
 25 which provides high strength, high corrosion resistance and excellent brazability, without being lowered by brazing process without noncorrosive flux, and is suited for the member used for assembling a high-strength heat exchanger.

To achieve the above object, the inventors hereof made studies on members for brazing which have a smaller amount of Mg contained in the aluminum alloy for the core member but have high strength while
 30 brazability and corrosion resistance are still more improved. As a result, the present invention is accomplished as follows.

An aluminum alloy composite material for brazing according to the present invention comprises a core member made of an aluminum alloy, which consists essentially of 0.2 to 1.5 wt.% of Mn, 0.3 to 1.3 wt.% of Si, 0.02 to 0.3 wt.% of Ti, the content of Mg being restricted to 0.2 wt.% or less as an impurity, and, as
 35 required, 0.6 wt.% or less of Cu, 0.3 wt.% or less of Cr and 0.2 wt.% or less of Zr being contained; an Al-Si filler member clad on one surface of said core member; and a cladding member of an aluminum alloy clad on the opposite side of said core member, which consists essentially of 0.3 to 3 wt.% of Mg and, as required, 5 wt.% or less of Zn being contained.

The present invention provides a high-strength aluminum alloy composite material which is not lowered
 40 in brazability or corrosion resistance by Nocolok brazing method. Accordingly, when the high-strength aluminum alloy composite material of the present invention is used for assembling a heat exchanger of an automobile, it provides remarkable effects such as a reduction in thickness and weight of the heat exchanger, reduction in the cost, and the like.

Fig.1 is a view showing a structure of an example of an aluminum alloy composite material for brazing,
 45 according to the present invention; and

Fig.2 is a view showing a structure of an application example of an aluminum alloy composite material for brazing, according to the present invention.

The present invention will be hereinafter described in detail.

The structure of an aluminum alloy composite material for brazing according to the present invention is
 50 such that, as shown in Fig.1, a core member 1, made of an aluminum alloy, the content of Mg being restricted to 0.2 wt.% or less (preferably 0.1 wt.% or less) as an impurity, consisting essentially of 0.2 to 1.5 wt.% of Mn, 0.3 to 1.3 wt.% of Si, 0.02 to 0.3 wt.% of Ti, and containing further as required, 0.6 wt.% or less of Cu, 0.3 wt.% or less of Cr and 0.2 wt.% or less of Zr, contacts an Al-Si filler member 3; and a cladding member 2 of aluminum alloy consisting essentially of a predetermined amount of (0.3 to 3 wt.%)
 55 of Mg is clad on the opposite surface of the core member 1.

The aluminum alloy composite material with the above structure has the following characteristics and thus achieves the intended effects.

In brazing, the brazability of the composite material is never lowered since the Mg content of the core

member 1 is restricted to a low value. Further, during brazing with heat, Si in the filler member 3 is diffused into the core member 1, while Mg is diffused into the core member 1 from the cladding member 2 on the opposite side of the core member 1, whereby MgSi compound is produced in the core member 1, thus increasing the post-brazing strength. In this case, the region of the core member 1 contacting the filler member 3 has the lowest Mg content since the Mg is supplied to the core member 1 from the cladding member 2 clad on the core member 1 on the opposite side of the filler member 3 and diffuses in the core member 1 to said region, whereby the strength can be improved without a reduction in the brazability. On the other hand, the surface of the cladding member 2 at the opposite side has the highest Mg content and the lowest Si content, and accordingly, the corrosion resistance is hardly lowered.

According to the aluminum alloy composite material proposed in Japanese Published Unexamined Patent Application No.175093/1990 which is a prior art related to the present invention, Ti content of the composition for the core member was not regulated, so that the corrosion resistance was found out to be insufficient.

Meanwhile, the core member of the present invention is added as essential components with Mn, Si and Ti, and this is because these alloy elements are indispensable for improving the strength and corrosion resistance (or inhibition of intergranular corrosion sensitivity) of core member. As required the strength as well as outside corrosion resistance can be further increased by adding Cu, while suitable amounts of Cr, Zr and the like regulate crystalline grain shape of the core member, thus improving the brazability. The addition of Mg within a regulated range also works as an inhibitor for reduction in brazability and contributes to improve the strength.

For the cladding member 2 clad on the opposite side of the core member 1, an Aluminum alloy containing Mg is used, which can function as a sacrificial anode when used for the radiator tube, thereby remarkably increasing corrosion resistance. By adding a suitable amount of Zn to the cladding member 2, the function of the sacrificial anode can further be improved.

Thus, during brazing, the core member 1 serves to restrain the diffusion of Si from the filler member 3 clad on one surface of the core member 1 and the diffusion of Mg from the cladding member 2 clad on the opposite surface of the core member 1. The thickness of the core member 1 will be optimum when the aforementioned effect can be obtained simultaneously with the strength being improved to its maximum.

The aluminum alloy composite material according to the invention has the above-mentioned effects.
Now, the definition of the contents of the individual elements will be described.

Core Member:

In the aluminum alloy for a core member, the allowable amount of Mg contained in aluminum alloy core member 1 as impurity before brazing is defined as 0.2 wt.% at the maximum, to avoid the brazability being lowered. If the Mg content is greater than the allowable value, the brazability is unfavorably lowered especially during a brazing process by a Nocolok brazing method. The Mg content is preferably less than 0.1wt.% or less.

By adding Mn to the core member 1, the corrosion resistance, brazability and strength can be improved. The adding amount of Mn for this purpose is to fall within a range of 0.2 to 1.5 wt.%. If the addition of Mn is less than 0.2 wt.%, the above advantages are not fully achieved, whereas the Mn content is greater than 1.5 wt.%, large compounds are produced and thus the workability is lowered.

Similarly, by adding Si to the core member 1, it is possible to improve the strength even if the supply of Si from the filler member 3 is insufficient. A preferable amount of Si to be added is 0.3 wt.% to 1.3 wt.%. If the amount of Si is smaller than 0.3 wt.%, the strength is not sufficiently improved, and if the amount of Si is greater than 1.3 wt.%, a disadvantage arises because the melting point is lowered.

Moreover, addition of Ti to the core member 1 can further improve the corrosion resistance. A preferable content of Ti to be added is 0.02 to 0.3 wt.%. If the content of Ti is less than 0.02 wt.%, the improvement of the corrosion resistance can not be achieved, and if the content of Ti exceeds 0.3 wt.%, the effect will reach saturation, and production of large compounds arises, thus disadvantageously resulting in low workability. As mentioned above, the Ti is indispensable for improving the corrosion resistance. This mechanism is performed such that Ti precipitates to form Ti-rich layers in the core member 1 in a rolling direction, thereby to inhibit a pitting corrosion from propagating toward the depth, and Ti makes the electrical potential of the core member 1 change toward positive. Since Ti diffuses at a slow speed in an aluminum alloy and moves not so far during brazing, the potential difference between the core member 1 and the filler member 3, or between the core member 1 and the cladding member 2 can be maintained, and thus the core member 1 can be effectively electrochemically protected from the corrosion.

Furthermore, adding Cu, as required, to the core member 1 makes the potential of the core member 1

transfer toward positive, and thus electrochemically protecting the core member 1 from the corrosion. For this reason, the addition of Cu causes a great effect for improving the corrosion resistance and the strength. A preferable amount of Cu is 0.6 wt.% or less. In this case, if, particularly, the content of Cu is 0.1wt.% or less, it is advantageous for inhibiting the sensitivity of the intergranular corrosion, and if the content of Cu is 5 0.3 to 0.6 wt.%, the strength is effectively improved. If, however, the amount of Cu exceeds 0.6 wt.%, although the strength is improved, the corrosion resistance is greatly lowered. As a result, it is impossible to ensure a sufficient corrosion resistance even though the function of the cladding member 2 as a sacrificial anode is enhanced, and thus brazability is lowered.

Similarly, adding at least one element selected form Cr and Zr to the core member 1 makes it possible 10 to improve the corrosion resistance, strength and brazability. Preferable amounts of Cr and Zr are 0.3 wt.% or less and 0.2 wt.%, respectively. Addition of the elements in excess of their respective upper limits leads to saturation of the aforementioned effects and decreases in workability.

The core member 1 serves to restrict not only the diffusion of Si from the filler member 3 but the 15 diffusion of Mg from the cladding member 2 during brazing. The thickness of the core member 1 is set to an optimum value such that the above-mentioned effects are achieved and the strength is the maximum, and preferably is 2.5 times or more the thickness of the filler member 3 and in the range of 0.1 to 1 mm.

Cladding member:

20 For the cladding member 2 clad on the surface of the core member 1 at the opposite side of the filler member 3, an aluminum alloy containing Mg is used, which can act as a sacrificial anode when the composite material is used for a radiator tube and the like, thereby remarkably increasing the corrosion resistance. The Mg content at the interface between the core member 1 and the filler member 3 after brazing is preferably about 0.1 wt.% to 0.2 wt.%, and to obtain such Mg content, the Mg content of the 25 cladding member 2 should be in the range of 0.3 wt.% to 3.0 wt.%, though the content depends on the thickness of the cladding member 2, the brazing conditions, and the like. If the Mg content is less than 0.3 wt.%, the improvement in strength is insufficient, and if the Mg content is greater than 3.0 wt.% then it is difficult to have the cladding member 2 clad on the core member 1. If a composite material with an increased thickness is to be manufactured, Mn, Cu, Cr, Ti, Zr, and the like may be added to the cladding 30 member 2 to improve the strength. The preferable amounts of these elements to be added are the same as in the case of adding the elements to the alloy of the core member 1.

When the cladding member 2 is to be used as a sacrificial anode, it is effective to add elements for transferring the potential toward positive, e.g., Mn, Cu, Ti, Cr, and the like, to the core member 1; in addition, to increase anode potential of the cladding member 2, Zn may be added to the cladding member 35 2. Preferably, the amount of Zn to be added is 5 wt.% or less, and if the cladding member 2 contains more Zn, a disadvantage can be caused in that the furnace is contaminated during brazing.

Filler Member:

40 For a filler member 3, Al-Si alloys such as, for instance, A4045 alloy and the like can be used as in the conventional art. It is also possible to add Zn to the filler member 3 in order to function the filler member 3 as a sacrificial anode. Preferable amount of Zn to be added to the filler member 3 is the same as in the case of adding the elements to the alloy of the cladding member 2.

Other Conditions:

As for the thickness of the composite material, 0.4 mm or less is enough when the material is used for a tube member of a radiator, because the strength is sufficiently increased by the diffusion of Mg from the cladding member 2 clad on the side of the core member 1 which is opposite to the filler member 3.

50 The present invention is the most effective when applied to the brazing method with noncorrosive flux which can extremely degrade the brazability due to Mg, but of course, it can be applied to other brazing methods such as atmosphere brazing, flux brazing, vacuum brazing, and the like. If, as shown in Fig.2, aluminum alloy core members 1 containing no Mg are clad on both surfaces of an aluminum alloy cladding member 2 containing Mg and Al-Si filler members 3 are clad on both surfaces of the resultant structure, then a brazing sheet with filler members 3 on both surfaces thereof can be obtained.

In assembling a radiator tube, the composite material is worked into a tube by a suitable process, e.g., by seam welding, with the aforesaid filler member 3 outside.

For the improvement in the corrosion resistance of the inner surface of the radiator tube, the most

effective measure is to use the cladding member 2 as a sacrificial anode for the core member 1. Since the brazing method with noncorrosive flux is carried out under atmospheric pressure, Zn scarcely vaporizes but is diffused into the core member 1 by the brazing heat, thus deceasing the surface concentration. To greatly improve the corrosion resistance of the inner surface (coolant-side surface) of the radiator tube by the sacrificial anode effect of the cladding member 2, the difference of pitting potentials between the surface of the cladding member 2 and the core member 1 need be not smaller than 30 mV. However, if the difference of the pitting potentials is greater than 120 mV, the speed of consumption of the cladding member 2 is large and thus the sacrificial anode effect cannot be maintained for a long period of time. The difference of the pitting potentials between the cladding member 2 and the core member 1 varies depending on the composition of the cladding member 2 and core member 1 and the brazing conditions, but if the pitting potential difference after the brazing with noncorrosive flux is within the range of 30 to 120 mV, excellent corrosion resistance can be enjoyed for a long time. Similarly, if the pitting potential difference between the surface of the filler member 3 and the core member 1 falls within the range of 30 to 120 mV, it is possible to ensure excellent corrosion resistance for a long period of time.

15 Embodiments of the present invention will now be described.

(Embodiment 1)

For examples 1 to 5 hereinafter, aluminum alloys (core member, cladding member and filler member) having the chemical compositions shown in Table 1 were prepared. In Table 1, No.1 to No.12 alloys were used for the core member, No.13 to No.19 alloys for the cladding member clad on the opposite side of the core member, and No.20 alloy (4045 alloy) for the filler member.

20 Here, No.7 to No.12 alloys are used as comparative examples, and No.7 alloy contained a large amount of Ti, and No.8 alloy contained little amount of Si and Ti. No.9, No.10, No.11 and No.12 alloys contained Mn, Si, Mg and Cu in a large amount respectively.

25 These aluminum alloys were combined as shown in Table 2, to produce samples of aluminum alloy composite materials for brazing having the structure shown in Fig.1.

Example 1-1:

30 Each of the aluminum alloy composite material for brazing having a respective combination shown in Table.2 was applied with 5g/m² brazing flux of noncorrosive on the filler member thereof, dried and then heated at 600 °C for 5 minutes in a nitrogen gas atmosphere having a dew point of -40 °C. The evaluation results on brazing for each are shown in Table.3. Here, the brazability of the samples was evaluated in terms of flow factor. As is apparent from Table.3, the examples according to the present invention had improved brazability.

Example 1-2:

40 The aluminum alloy composite materials heated under the same conditions as in Example 1 were left to stand for 7 days in room temperature, and then subjected to a tensile test. As seen from Table 3 showing the results, the examples according to the present invention had high strength after brazing over 15 kgf/mm²(145N/mm²)

Example 1-3:

45 The aluminum alloy composite materials heated under the same conditions as in Example 1-1 were subjected to a CASS test to evaluate the corrosion resistance of the filler member side of the composite material. The evaluation results after the CASS test for 250 hours are shown in Table 3. As indicated in Table.3 under the caption "Corrosion depth of Filler Member Side (mm)", the examples according to the invention had excellent corrosion resistance.

Example 1-4:

50 The aluminum alloy composite materials heated under the same conditions as in Example 1-1 were subjected to an immersion test by immersing the composite materials in artificial water (Cl⁻: 300ppm; SO²⁻: 100ppm; Cu⁺: 5ppm) for 30 days under alternate temperature conditions of 88 °C for 8 hours and room temperature for 16 hours, to evaluate the corrosion resistance of the cladding member clad on the

core member. The results of the immersion test are shown in Table.3. As indicated in Table.3 under the caption "Corrosion depth of Cladding Member Side (mm)", the examples according to the invention had excellent corrosion resistance.

5 Example 1-5:

For each of the aluminum alloy composite materials heated under the same conditions as in Example 1-1, the pitting potential of the surface of the cladding member clad on the core member, the pitting potential of the surface of the filler member and the pitting potential of the core member which was exposed by 10 mechanically removing the cladding member with emery grinding paper were measured under the following conditions.

<Measuring Conditions:>

15 Electrolyte: 3.5% NaCl, (deaeration)
 Temperature: 25 °C
 Measuring Method: Potential scanning by using a potentiostat (sweep speed: 10 mV/min)

The measured differences of pitting potentials between the core member and the cladding member are shown in Table.3. As shown in Table.3, the differences of pitting potentials between the core member and 20 cladding member and between the core member and the filler member, of the aluminum alloy composite materials for brazing according to the present invention fall within the range of 30 to 120 mV, and this means that excellent corrosion resistance can be maintained for a long period of time.

(Embodiment 2)

25 For examples 2-1 to 2-5 hereinafter, aluminum alloys (core member, cladding member and filler member) having the chemical compositions shown in Table.4 were prepared. In Table 4, No.1 to No.12 alloys were used for the core member, No.13 to No.19 alloys for the cladding member, and No.20 alloy (4045 alloy) for the filler member.

30 Here, No.7 to No.12 alloys are used as comparative examples, and No. 7 alloy contained a large amount of Ti, and No.8 alloy contained little amount of Si and Ti. No.9, No.10, No.11 and No.12 alloys contained Mn, Si, Mg and Cu in a large amount, respectively.

These aluminum alloys were combined as shown in Table 5, to produce samples of aluminum alloy composite materials for brazing having the structure shown in Fig.1.

35 Example 2-1:

Each of the aluminum alloy composite material for brazing having a respective combination shown in Table.5 was applied with 5g/m² Nocolok brazing flux on the filler member thereof, dried and then heated at 40 600 °C for 5 minutes in a nitrogen gas atmosphere having a dew point of -40 °C. The evaluation results on brazing for each are shown in Table.6. Here, the brazability of the samples was evaluated in terms of flow factor. As is apparent from Table.6, the examples according to the present invention had improved brazability.

45 Example 2-2:

The aluminum alloy composite materials heated under the same conditions as in Example 2-1 were left to stand for 7 days in room temperature, and then subjected to a tensile test. As seen from Table.6 showing the results, the examples according to the present invention had high strength after brasing over 15 50 kgf/mm² (145 N/mm²).

Example 2-3:

The aluminum alloy composite materials heated under the same conditions as in Example 2-1 were 55 subjected to a CASS test to evaluate the corrosion resistance of the filler member side of the composite material. The evaluation results 250 hours after the CASS test are shown in Table 6. As indicated in Table.6 under the caption "Corrosion depth of Filler Member Side (mm)", the examples according to the present invention had excellent corrosion resistance.

Example 2-4:

The aluminum alloy composite materials heated under the same conditions as in Example 2-1 were subjected to an immersion test by immersing the composite materials in artificial water (Cl^- : 300ppm; 5 SO_4^{2-} : 100ppm; Cu^+ : 5ppm) for 30 days under alternate temperature conditions of 88°C for 8 hours and room temperature for 16 hours, to evaluate the corrosion resistance of the cladding member clad on the core member. The results of the immersion test are shown in Table.6. As indicated in Table.6 under the caption "Corrosion depth of Cladding Member Side (mm)", the examples according to the present invention had excellent corrosion resistance.

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Example 2-5:

For each of the aluminum alloy composite materials heated under the same conditions as in Example 2-1, the pitting potential of the surface of the cladding member, the pitting potential of the surface of the filler member and the pitting potential of the core member which was exposed by mechanically removing the cladding member with emery grinding paper were measured under the same conditions as in Example 1-5 of Embodiment 1.

15 The measured differences of pitting potentials between the core member and the cladding member are shown in Table.6. As shown in Table.6, the differences of pitting potentials between the core member and cladding member and between the core member and the filler member, of the aluminum alloy composite materials for brazing according to the present invention fall within the range of 30 to 120 mV, and this means that excellent corrosion resistance can be maintained for a long period of time.

20 As is apparent from Embodiments 2-1 and 2-2, each of the examples of the present invention can provide high strength without a reduction in brazability or corrosion resistance.

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Table 1
Chemical Composition of Aluminum Alloy
used in Aluminum Alloy Composite Material for Brazeing

Alloy No.	Chemical Composition (wt. %)								Al
	Si	Cu	Mn	Mg	Zn	Cr	Zr	Ti	
1	0.8	0.0	1.2	0.00	0.0	0.10	0.05	0.10	bal
2	0.8	0.0	1.5	0.00	0.0	0.05	0.00	0.15	Core Member Example
3	1.2	0.0	1.0	0.00	0.0	0.10	0.05	0.10	Ditto
4	0.8	0.0	1.0	0.10	0.0	0.00	0.15	0.15	Ditto
5	0.8	0.1	0.4	0.00	0.0	0.10	0.05	0.05	Ditto
6	0.8	0.0	1.0	0.00	0.0	0.00	0.00	0.15	Ditto
7	0.8	0.0	1.0	0.00	0.0	0.10	0.05	0.35	Core Member Comparative Example
8	0.2	0.1	1.0	0.00	0.0	0.10	0.05	0.01	Ditto
9	0.8	0.2	1.6	0.00	0.0	0.10	0.05	0.10	Ditto
10	1.4	0.0	0.2	0.00	0.0	0.10	0.05	0.10	Ditto
11	0.8	0.1	1.0	0.25	0.0	0.10	0.05	0.05	Ditto
12	0.8	0.7	1.0	0.00	0.0	0.10	0.05	0.15	Ditto
13	-	-	-	1.0	4.5	-	-	-	Cladding Member Example
14	-	-	-	2.0	2.0	-	-	-	Ditto
15	-	-	-	3.0	0.0	-	-	-	Ditto
16	-	-	-	0.5	1.5	-	-	-	Ditto
17	-	-	-	1.0	2.0	-	-	-	Ditto
18	-	-	-	0.0	1.5	-	-	-	Ditto
19	10.5	0.0	-	4.0	1.0	-	-	-	Cladding Member Comparative Example
20	10.5	0.0	1.0	0.0	0.0	0.10	0.05	0.10	Ditto Filler Member :4045

bal: balance

Table.2
Structure of Aluminum Alloy Composite Material for Brazeing

Test No.	Core Member		Cladding Member		Filler Member		Total Thick-ness of sheet (mm)	Reference
	No.	Thick-ness (mm)	No.	Thick-ness (mm)	No.	Thick-ness (mm)		
1	1	0.24	14	0.03	20	0.03	0.30	Example
2	2	0.24	14	0.03	20	0.03	0.30	Ditto
3	3	0.24	14	0.03	20	0.03	0.30	Ditto
4	4	0.24	14	0.03	20	0.03	0.30	Ditto
5	5	0.24	14	0.03	20	0.03	0.30	Ditto
6	6	0.24	14	0.03	20	0.03	0.30	Ditto
7	7	0.24	14	0.03	20	0.03	0.30	Comparative Example
8	8	0.24	14	0.03	20	0.03	0.30	Ditto
9	9	0.24	14	0.03	20	0.03	0.30	Ditto
10	10	0.24	14	0.03	20	0.03	0.30	Ditto
11	11	0.24	14	0.03	20	0.03	0.30	Ditto
12	12	0.24	14	0.03	20	0.03	0.30	Ditto
13	13	0.24	13	0.03	20	0.03	0.30	Example
14	14	0.24	15	0.03	20	0.03	0.30	Ditto
15	15	0.24	16	0.03	20	0.03	0.30	Ditto
16	16	0.24	17	0.03	20	0.03	0.30	Ditto
17	17	0.24	18	0.03	20	0.03	0.30	Comparative Example
18	18	0.24	19	0.03	20	0.03	0.30	Ditto
19	19	0.05	14	0.22	20	0.01	0.30	Ditto
20	20	1	0.08	14	0.01	20	0.15	0.10
21	21	1	1.2	14	0.15	20	0.02	1.5
22	22	1	0.16	14	0.02	20	0.02	0.20
23	23	1	0.32	14	0.04	20	0.04	0.40

Note: The No. in the columns for core member, cladding member and filler member indicate the alloy No. in Table. 1.

Table.3 Testing Result of Evaluation on Aluminum Alloy Composite Materials

Test No.	Brazing -bility (Flow Factor) (%)	Tensile Strength (N/mm ²)	Corrosion of Filler Member Side Corrosion Depth(mm)	Corrosion of Cladding Member Side Corrosion Depth(mm)	Cladding Member /Core Member Potential Difference (mV vs sec)	
1	70	165	0.08	0.03	70	Example
2	70	165	0.08	0.03	70	Ditto
3	70	175	0.15	0.03	70	Ditto
4	60	160	0.10	0.03	70	Ditto
5	70	170	0.15	0.03	80	Ditto
6	70	160	0.08	0.03	70	Ditto
7	70	165	Through hole	0.20	70	Comparative Example
8	65	110	0.10	0.03	70	Ditto
9	60	175	Through hole	0.20	80	Ditto
10	No evaluation can be made due to the melt of core member during brazing					
11	35	180	Through hole	0.20	20	Ditto
12	60	210	0.05	Through hole	110	Ditto
13	70	150	0.08	0.05	110	Example
14	70	180	0.08	0.10	30	Ditto
15	70	140	0.08	0.06	60	Ditto
16	70	150	0.08	0.05	80	Ditto
17	70	130	0.10	0.05	70	Comparative Example
18	No evaluation can be made due to the failure of cladding					
19	40	180	Through hole	Through hole	Unmeasurable	
20	50	180	Through hole	Through hole	Unmeasurable	
21	80	140	0.20	0.10	130	Ditto
22	60	185	0.08	0.03	60	Example
23	75	155	0.10	0.04	90	Ditto

Table.4
Chemical Composition of Aluminum Alloy
used in Aluminum Alloy Composite Material for Brazeing

Alloy No.	Chemical Composition (wt.-%)								Core Member Example	
	Si	Cu	Mn	Mg	Zn	Cr	Zr	Ti	Al	
1	0.3	0.4	0.6	0.05	0.0	0.25	0.05	0.10	bal	Core Member Example
2	0.8	0.3	1.5	0.00	0.0	0.05	0.00	0.15	bal	Ditto
3	0.8	0.5	1.0	0.00	0.0	0.10	0.05	0.10	bal	Ditto
4	1.2	0.5	1.0	0.20	0.0	0.00	0.15	0.15	bal	Ditto
5	0.8	0.6	0.4	0.00	0.0	0.10	0.05	0.05	bal	Ditto
6	0.8	0.5	1.0	0.00	0.0	0.00	0.00	0.15	bal	Ditto
7	0.8	0.5	1.0	0.00	0.0	0.10	0.05	0.35	bal	Core Member Comparative Example
8	0.2	0.1	1.0	0.00	0.0	0.10	0.05	0.01	bal	Ditto
9	0.8	0.2	1.6	0.00	0.0	0.10	0.05	0.10	bal	Ditto
10	1.4	0.0	0.2	0.00	0.0	0.10	0.05	0.10	bal	Ditto
11	0.8	0.1	1.0	0.25	0.0	0.10	0.05	0.05	bal	Ditto
12	0.8	0.7	1.0	0.00	0.0	0.10	0.05	0.15	bal	Ditto
13	-	-	1.0	4.5	-	-	-	-	bal	Cladding Member Example
14	-	-	-	2.0	2.0	-	-	-	bal	Ditto
15	-	-	-	3.0	0.0	-	-	-	bal	Ditto
16	-	-	-	0.5	1.5	-	-	-	bal	Ditto
17	-	-	-	1.0	2.0	-	-	-	bal	Ditto
18	-	-	-	0.0	1.5	-	-	-	bal	Ditto
19	-	0.0	1.0	4.0	1.0	-	-	-	bal	Cladding Member Comparative Example
20	10.5	-	-	0.0	0.0	0.10	0.05	0.10	bal	Ditto Filler Member :4045

bal: balance

Table 5
Structure of Aluminum Alloy Composite Material for Brazeing

Test No.	Core Member		Cladding Member		No. Filler Member	Total Thick-ness of sheet (mm)	Reference
	No.	Thick-ness (mm)	No.	Thick-ness (mm)			
1	1	0.24	14	0.03	20	0.03	0.30 Example
2	2	0.24	14	0.03	20	0.03	0.30 Ditto
3	3	0.24	14	0.03	20	0.03	0.30 Ditto
4	4	0.24	14	0.03	20	0.03	0.30 Ditto
5	5	0.24	14	0.03	20	0.03	0.30 Ditto
6	6	0.24	14	0.03	20	0.03	0.30 Ditto
7	7	0.24	14	0.03	20	0.03	0.30 Comparative Example
8	8	0.24	14	0.03	20	0.03	0.30 Ditto
9	9	0.24	14	0.03	20	0.03	0.30 Ditto
10	10	0.24	14	0.03	20	0.03	0.30 Ditto
11	11	0.24	14	0.03	20	0.03	0.30 Ditto
12	12	0.24	14	0.03	20	0.03	0.30 Ditto
13	13	0.24	13	0.03	20	0.03	0.30 Example
14	14	0.24	15	0.03	20	0.03	0.30 Ditto
15	15	0.24	16	0.03	20	0.03	0.30 Ditto
16	16	0.24	17	0.03	20	0.03	0.30 Ditto
17	17	0.24	18	0.03	20	0.03	0.30 Comparative Example
18	3	0.24	19	0.03	20	0.03	0.30 Ditto
19	3	0.05	14	0.22	20	0.03	0.30 Ditto
20	3	0.08	14	0.01	20	0.01	0.10 Ditto
21	3	1.2	14	0.15	20	0.15	1.5 Ditto
22	3	0.16	14	0.02	20	0.02	0.20 Example
23	3	0.32	14	0.04	20	0.04	Ditto

Note: The No. in the columns for core member, cladding member and filler member indicates the alloy No. in Table 4.

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Table 6 Testing Result of Evaluation on Aluminum Alloy Composite Materials

Test No.	Brazability (Flow Factor) (%)	Tensile Strength (N/mm ²)	Corrosion of Filler Member Side Corrosion Depth(mm)	Corrosion of Cladding Member Side Corrosion Depth(mm)	Cladding Member /Core Member Potential Difference (mV vs sec)	
1	65	165	0.06	0.10	80	Example
2	65	195	0.06	0.15	70	Ditto
3	70	210	0.10	0.15	70	Ditto
4	55	230	0.08	0.15	80	Ditto
5	60	210	0.10	0.15	80	Ditto
6	70	200	0.05	0.15	80	Ditto
7	70	200	Through hole	0.20	70	Comparative Example
8	65	110	0.10	0.03	70	Ditto
9	60	175	Through hole	0.20	80	Ditto
10	No evaluation can be made due to the melt of core member during brazing	180	Through hole	0.20	20	Ditto
11	35	210	0.05	Through hole	120	Ditto
12	60	185	0.05	0.05	100	Example
13	70	210	0.05	0.15	40	Ditto
14	70	170	0.05	0.10	60	Ditto
15	70	180	0.05	0.05	80	Ditto
16	70	150	0.10	0.05	70	Comparative Example
17	No evaluation can be made due to the failure of cladding	180	Through hole	Through hole	Unmeasurable	
18		220	Through hole	Through hole	Unmeasurable	
19	40	160	0.20	0.15	130	
20	50	225	0.05	0.10	60	
21	80	200	0.06	0.04	90	
22	60					
23	75					

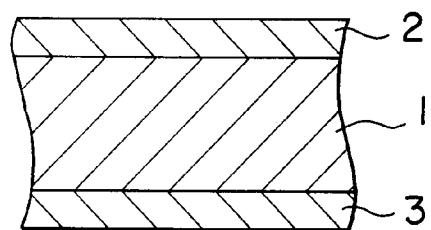
The composite material has a structure comprising: a core member(1) of an aluminum alloy, the content of Mg being restricted to 0.2 wt.% or less as an impurity, consisting essentially of 0.2 to 1.5 wt.% of Mn, 0.3 to 1.3 wt.% of Si, 0.02 to 0.3 wt.% of Ti, and, as required, 0.6 wt.% or less of Cu, 0.3 wt.% or less of Cr and 0.2 wt.% or less of Zr; an Al-Si filler member(3) clad on one surface of the core member; and a cladding member(2) of an aluminum alloy clad on the opposite side of said core member 1, consisting essentially of

0.3 to 3 wt.% of Mg and as required, 5 wt.% or less of Zn. The thickness of the core member(1) is preferably, 2.5 times or more greater than that of the filler member(3) falling within a range of 0.1 to 1 mm. With respect to a tube member for a radiator which is assembled by the brazing method with noncorrosive flux, the pitting potential of the core member (positive) is higher than that of the cladding member and is
5 higher than that of the filler member, and the differences in pitting potential between the core member and the cladding member and between the core member and the filler member are 30 to 120 mV. With this, it is possible to obtain a tube member for a radiator which has excellent corrosion resistance for a long period of time.

10 Claims

1. An aluminum alloy composite material for brazing comprising a core member(1), an Al-Si filler member (3) clad on one surface of the said core member and a cladding member(2) of an aluminum alloy clad on the opposite side of said core member, characterized in that
15 said core member(1) is made of an aluminum alloy consisting essentially of 0.2 to 1.5 wt.% of Mn, 0.3 to 1.3 wt.% of Si, 0.02 to 0.3 wt.% of Ti, the content of Mg being restricted to 0.2 wt.% or less as an impurity; and
 said cladding member(2) is made of an aluminum alloy consisting essentially of 0.3 to 3 wt.% of Mg.
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2. The aluminum alloy composite material for brazing according to claim 1, characterized in that said core member(1) further contains 0.6 wt.% or less of Cu, 0.3 wt.% or less of Cr and 0.2 wt.% or less of Zr.
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3. The aluminum alloy composite material for brazing according to claim 1 or 2, characterized in that said cladding member(2) further contains 5 wt.% or less of Zn.
4. The aluminum alloy composite material for brazing according to claim 2, characterized in that the Cu content of said core member(1) is 0.1 wt.% or less.
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5. The aluminum alloy composite material for brazing according to claim 2, characterized in that the Cu content of said core member(1) is 0.3 to 0.6 wt.%.
6. The aluminum alloy composite material for brazing according to claim 1, characterized in that said core member(1) has a thickness 2.5 times or more greater than that of said filler member(3) and falling
35 within a range of 0.1 to 1mm.
7. A tube member in which said aluminum alloy composite material for brazing according to claim 1 is fabricated into a tube with said filler member(3) outside, by seam welding.
8. The tube member for a radiator according to claim 7, which is assembled by brazing method with noncorrosive flux.
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9. The tube member for a radiator according to claim 8, characterized in that, after the brazing process, the pitting potential of said core member(1) (positive) is higher than that of said cladding member(2) and is higher than that of said filler member(3), and the differences in pitting potential between said core member and said cladding member and between said core member and said filler member fall 30 to 120 mV.
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F I G. 1



F I G. 2

